Contribution to Magnetic Anisotropy from Cations Which Locally Distort the Crystal Lattice

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The energy states of the Ni²⁺ ion on the tetrahedral sites of the spinel lattice have been calculated as a function of tetragonal distortions from cubic symmetry. The computed first- and second-order contributions from tetrahedral Ni²⁺ to the cubic anisotropy of magnetic spinels have been compared with experiment. Less than 0.1% tetrahedral Ni²⁺ could be responsible for the anisotropy of NiFe₂O₄. Computed values for the zero-field splitting of Cr³⁺ in MgAl₂O₄ and the anisotropy of octahedral Co²⁺ in magnetic spinels are consistent with experiment for a trigonal field 10% of the cubic field in strength and with a sign corresponding to that due to the nearest neighbor oxygen ions. The single-ion first- and second-order contributions to the cubic anisotropy from octahedral Ni³⁺ have also calculated and found to be very large and positive.

1. Introduction

The crystalline electric fields, experienced by the magnetic ions in oxides, play a major role in determining the contribution such ions have to the net magnetic anisotropy. Calculations on single-ion contributions to the magnetic anisotropy have heretofore been restricted to certain ions which permit great simplifications in the theory. However, little concern has been expressed for the many ions which could not be considered before and yet which should have very large contributions to the magnetic anisotropy. This paper is a preliminary report on work directed toward the calculation of the single-ion contributions to magnetic anisotropy for the transition metal ions including those ions which tend to distort their local environment. The method of treatment is first outlined and the some of the more important results obtained for d^2 , d^3 , d^7 and d^8 ions are discussed.

2. Method of Treatment

A. The Isolated Ion

The first step in the anisotropy calculation is to obtain the electronic states of the ion in the crystalline electric field as if the ion were magnetically isolated from the lattice. The magnetic exchange is subsequently introduced as a perturbation.

The Hamiltonian, H_0 , for the isolated d^N or d^{10-N} ion is

$$H_0 = \sum_{i,$$

where the V_i are the crystalline electric field

potentials for each electron. The last term of H_0 can be conveniently expressed in terms of the tensor operator, T_{i^m} , which operate directly on the space coordinate of the many electron wave function;

$$T_{i}^{m} = \sqrt{\frac{4\pi}{2l+1}} Y_{i}^{m}, \qquad (2)$$

where the Y_{i}^{m} are the usual spherical harmonics. For example, an ion on a lattice site having tetragonal symmetry would have the potential:

$$\sum_{i} V_{i} = 21 Dq \left\{ T_{4}^{\circ} + \sqrt{\frac{5}{7}} \left(\frac{T_{4}^{+4} + T_{4}^{-4}}{\sqrt{2}} \right) \right\} \\ + 21 Qq \left\{ T_{4}^{\circ} - 3\sqrt{\frac{5}{7}} \left(\frac{T_{4}^{+4} + T_{4}^{-4}}{\sqrt{2}} \right) \right\} \\ + 210 Ap T_{2}^{\circ}, \qquad (3)$$

where the Dq, Qq, and Ap are the cubic and tetragonal crystalline field parameters. The first term of H_0 is the internal electron exchange of the ion, which can be expressed in terms of the Racah¹⁾ parameters *B* and *C*.

The electronic states (45 for d^2 or d^8 , and 120 for d^3 or d^7 ions) produced by the Hamiltonian H_0 can be obtained for any arbitrary set of parameters. The secular determinant is first factored according to the symmetry of the problem and then exact diagonalization is carried out for each factor. Parameters are so chosen as to be consistent with optical spectra measured for the particular ion in a similar diamagnetic lattice.

B. Magnetic Exchange

Once the electronic states for the isolated ion are found, one must include the magnetic exchange between the given ion and all the remaining magnetic ions in the crystal. At the present time this has been accomplished using the molecular field approximation. Thus, we add to the Hamiltanian H_0 the perturbation:

$$H_1 = 2\mu_B S \cdot H_e , \qquad (4)$$

where μ_B =Bohr magneton and H_e is the effective exchange field. The electronic states in the presence of exchange are then obtained by diagonalization of the secular determinant for the lowest manifold of states.

C. Anisotropy

The magnetic anisotropy energy is associated with the anisotropic part of the total free energy of the crystal. The total firstorder anisotropy constant for a cubic material can then be expressed²) as

$$K_1 = 4(F_{<110>} - F_{<100>}),$$
 (5)

where $F_{<110>}$ is the free energy of the magnetic crystal with the magnetization along a <110> direction, and similarly for $F_{<100>}$.

If now the given ion of interest is sufficiently dilute in the lattice, the net contribution of all such ions should then have the same form as Equation (5) except that the free energies indicated are the partial free energies for these given ions. For an example, let us consider the case of an ion which has tetragonal symmetry so that there is a unique "z" direction along one of the cubic axes, there are three such sets of ions in the cubic lattice each having a different unique axis. For each set of ions there will be a contribution having the form of Equation (5), and after summing over each of these sets one obtains:

$$K_{1}(T,H) = \frac{-4N}{3} \times kT \left\{ \ln\left(\frac{Z_{xz}}{Z_{z}}\right) + \ln\left(\frac{Z_{xz}}{Z_{x}}\right) + \ln\left(\frac{Z_{xy}}{Z_{x}}\right) \right\}, \quad (6)$$

where N is the total number of the given ions and $Z_{xy}(T, H)$ is the single-ion partition function for the given ion at the temperature T with an exchange field H in the xy direction, etc. Similar expressions can also be developed for the single-ion contribution to the secondorder anisotropy.

All phases of this treatment from the solution of the states of the isolated ion to the actual anisotropy calculation have been programmed for successive operation on the IBM 7090 computer.

3. The Tetrahedral Ni²⁺ Ion

The treatment described above has been applied to the d^2 , d^3 , d^7 , and d^8 ions (V³⁺, Cr³⁺, Ni⁸⁺, Co²⁺, and Ni²⁺). However, because of space limitations only one important case will be discussed in any detail; namely, that for the Ni²⁺ ion when it occupies the tetrahedral sites, or A sites, of the spinel lattice.

The Ni²⁺ ion on the A sites tends to distort its local environment from cubic to tetragonal symmetry. Therefore, the tetragonal crystalline field potential given in Equation (3) may be used. The lowest states of the isolated ion are given in Fig. 1, for the set of parameters; $B=1030 \text{ cm}^{-1}$, C=4.5B, $\lambda=-618 \text{ cm}^{-1}$

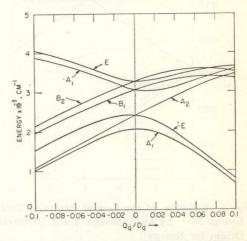


Fig. 1. Lowest States of Tetrahedral Ni^{2+} with Tetragonal Distortion (Symbols Refer to the Irreducible Representations of the D_{4h} Double Group).

 $Dq=412 \text{ cm}^{-1}$, Ap=-Qq which are consistent with the optical spectra. The values of the distortion parameters Qq/Dq=-0.1, 0, +0.1correspond to a local c/a=1.18, 1.0, 0.92, respectively. The energy of the ground state in Fig. 1 is lowered by about the same amount by both signs of the distortion parameter. Without a knowledge of the nuclear potential it is thus difficult to predict the sign of the distortion. The local distortions about the concentrated Ni²⁺ ions in the compound NiCr₂O₄ appear to be such that $c/a > 1^{3}$. Nevertheless, the local distortions about very dilute tetrahedral Ni²⁺ ions in a cubic structure could be either prolate or oblate. Since the Ni²⁺ ion has a unique "z" axis along one of the cubic axes of the crystal, we can use the expression for the anisotropy contribution given in Equation (6). For the case of $T=0^{\circ}$ K, Equation (6) reduces to a much simpler form for the Ni²⁺ ion:

$$K_{1}(0, H) = \frac{4N}{3} (2E_{xz}^{\circ} - E_{z}^{\circ} - E_{z}^{\circ}), \quad (7)$$

where $E_{xz}^{\circ}(H)$ is the ground state when the exchange field is in the *xz* direction, and the fact that, for this ion, $E_{xy}^{\circ} = E_x^{\circ}$ has been used to simplify the expression. In Fig. 2 is shown a typical example of how the exchange field splits the lower states of Ni²⁺ on the A-sites

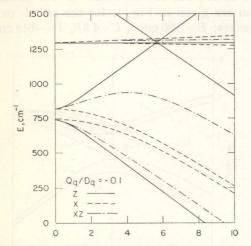


Fig. 2. Splitting of the Lowest States of Tetrahedral Ni²⁺ by an Exchange Field (Arbitrary Origin for Energy).

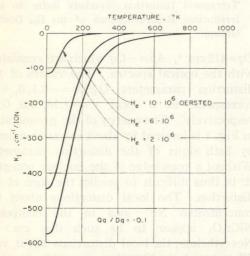


Fig. 3. Single-Ion Contribution of Tetrahedral Ni²⁺ to the First-Order Cubic Anisotropy.

of the spinel lattice. Using Equation (7) one can by inspection of Fig. 2 see qualitatively the behavior of the single-ion contribution to K_1 at T=0; K_1 is negative and large, increasing monotonically toward a limiting value at high fields.

Equation (6) and a similar one for the second-order anisotropy has been used to calculate the temperature dependence of the single-ion anisotropy contribution. The first and second order contribution for Ni^{2+} on A-sites with a prolate local distortion are given in Figs. 3 and 4. Similar results were obtained for Ni^{2+} on A-sites with a oblate local distortion except that the contribution to the second order anisotropy is positive and much smaller. These single-ion contributions are so large that a very small fraction of Ni^{2+} on the A-sites of magnetic spinels would have a profound influence on the anisotropy.

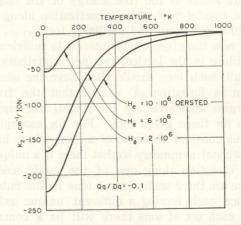


Fig. 4. Single-Ion Contribution of Tetrahedral Ni²⁺ to the Second-Order Cubic Anisotropy.

The magnetic anisotropy of NiFe₂O₄ is a case in point for which previous investigations⁴⁾ have entirely neglected the role of the Ni²⁺ on the A-sites. The measured⁵⁾ firstorder anisotropy, $K_1 = -0.04 \text{ cm}^{-1}/\text{molecule}$, has been previously attributed to the Fe³⁺ ion, but is more likely due to less than 0.1%of Ni²⁺ on the A-sites. Measurements^{5),6)} of the second-order anisotropy of NiFe₂O₄ vary in magnitude, but all are negative. The negative second order anisotropy would indicate that the influence of Ni²⁺ on prolately distorted A-sites predominates even though both types of local distortion may exist in this compound.

4. Octahedral d^3 and d^7 Ions

Other significant results have been computed for the case of the ions Cr^{3+} , Co^{2+} , and Ni³⁺ on the B-sites of the spinel lattice. The large zero field splitting observed⁷⁾ for Cr^{3+} in MgAl₂O₄ was found consistent with a trigonal field which is 10% of the cubic field strength and which has the sign associated with the field of the nearest neighbor oxygen ions. The anisotropy contribution from the Co^{2+} ion, in a trigonal field of the same sign as that for Cr^{3+} , was obtained and found consistent with experiment (for $H_e=4\cdot10^6$ oersted, $T=0^{\circ}$ K; $K_1=+48$ cm⁻¹/ion and $K_2=-150$ cm⁻¹/ ion). Similarly for Ni³⁺ the anisotropy contribution was found to be very large (for $H_e=4\cdot10^6$ oersted, $T=0^{\circ}$ K; $K_1=+80$ cm⁻¹/ion and $K_2=+800$ cm⁻¹/ion).

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DISCUSSION

S. KRUPIČKA: I would like to make a remark on the existence of local distortions in cubic spinels. About one year ago we attempted to get some evidence of the existence of such distortions by X-ray investigation. As there are some indications of the presence of some Mn³⁺ ions in stoichiometric manganese ferrite we determined the meansquare displacement of ions in this ferrite and compared it with m.s. displacements in other binary ferrites (Zn, Mg, Ni). It was found that displacements in Mn ferrite are greater about 0.06Å which was interpreted in agreement with a rough computation as due to Jahn-Teller distortions of Mn³⁺ octahedra (J. Phys. Chem. Solids 20 (1961), 167). More recently Mr. Cervinka from our Institute has made a refinement of that study using the monochromatic radiation. His results agree excellently with the previous ones; in addition, if we plot the logarithm of the integrated intensities (divided by $|F_{hkl}|^2$ and corrected for usual factors) against $\sin^2 \theta / \lambda^2$ we get two straight lines with different slopes each of which belongs to different systems of $\{hkl\}$. The slope of one line is the very same as for other ferrites studied but the others is pointing to a greater displacement of certain ions. So we believe that it will be possible to determine which ions are displaced by distortion.

P. K. BALTZER: I have no comment with regard to Dr. KRUPIČKA's remark. However, I would like to add that the reason the energy of the orbital singlet, for Ni²⁺ on A-sites, is lowered with an *oblate* distortion is that the second-order tetragonal crystalline field parameter Ap has the sign opposite to the fourth order tetragonal parameter Qq (this is opposite to that for ions on B sites).

K. SIRATORI: If Ni^{2+} on A site is the main cause of the magnetic anisotropy in $NiFe_2O_4$, we can expect that the magnitude of the anisotropy depends not only on oxidation or reduction of the sample, as was discovered by Mr. and Mrs. Sekizawa, but also on heat treatment even in a stoichiometric state. Do you know any experimental data on this point?

P. K. BALTZER: The dependence of the anisotropy of NiFe₂O₄ on the amount of Ni²⁺ on the 4-fold sites has not been investigated. If it is possible by proper heat treatment to vary the amount of Ni²⁺ on the A sites while keeping the chemistry invarient it should be possible to verify the dependence of the anisotropy of NiFe₂O₄ on tetrahedral Ni²⁺.