

INFRARED AND PHOTOEMISSION SPECTRA OF AMORPHOUS
FLUORINATED SILICON

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Si-F, Si-F₂, Si-F₃ and Si-F₄ species have been identified in a-Si:F,H films through their vibrational frequencies as measured with ir-spectroscopy. Upon annealing some of the Si-F transform into Si-F₄. Gas evolution experiments show that F is stable in a-Si:F up to the crystallization temperature. Fluorine 2p induced states between 9 and 14 eV below the valence band maximum have been identified by photoemission.

I. Introduction

Madan et al. [1] have suggested that the addition of fluorine to a-Si:H (a-Si:H,F) results in a material with a doping efficiency higher than that of the merely hydrogenated substance (a-Si:H). Furthermore, the room temperature conductivity of their phosphorus doped fluorinated material had a value of $\sim 1 \Omega^{-1} \text{ cm}^{-1}$, not previously obtained in amorphous hydrogenated silicon.

Here we report ir absorption and photoemission measurements on a-Si:H,F films. From these measurements we deduce the bonding configurations of F in the a-Si network. The photoemission measurements reveal directly the position of the fluorine induced states in the a-Si valence bands. The thermal stability of hydrogen and fluorine was investigated in annealing and differential gas evolution experiments.

II. The Preparation of Fluorinated a-Si

Films of fluorinated a-Si were prepared by RF sputtering from a c-Si target in a gas mixture containing argon, hydrogen, and silicon tetrafluoride (SiF₄) onto a substrate held at 65°C. The specific power level was $\sim 8.8 \text{ W/cm}^2$. A flow of argon gas was adjusted so as to maintain a pressure of 10^{-2} torr with the unthrottled pump. The flow rates of H₂ and SiF₄ were similarly adjusted to yield pressures in the range from 5×10^{-4} to 2×10^{-3} torr. The deposition rate obtained with these parameters was approximately $1 \mu\text{m/h}$. The samples investigated were between $1 \mu\text{m}$ and $2 \mu\text{m}$ thick.

III. Results and Discussion

Fig.(1) shows the infrared spectrum of an a-Si:H,F film obtained with a ratio of 5 for the SiF₄/H₂ partial pressures in the sputtering gas. The infrared absorption coefficient, $\alpha(\omega)$, was calculated from the transmission using the procedure of Brodsky et al. [2]. Table I summarizes the positions and assignments of peaks observed in the ir-spectra of 10 films prepared with different SiF₄/H₂ ratios.

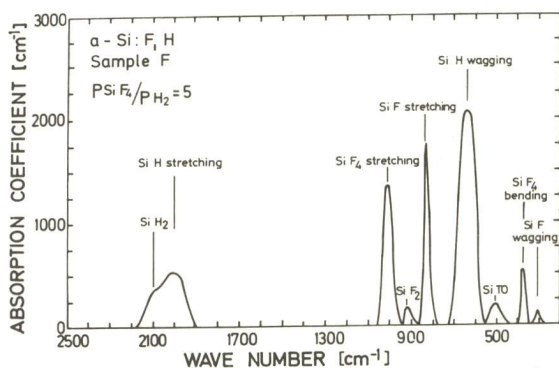


Fig. 1

The infrared spectrum of a-Si:F,H prepared at room temperature

brations of SiF_4 molecules imbedded in the a-Si matrix during the deposition process. The average Si-F bond-stretching frequency (longitudinal + transverse) in solid SiF_4 is 1013 cm^{-1} and the average bond-bending frequency is 385 cm^{-1} [3]. The cross sections of the two bands in crystalline SiF_4 [4] give a ratio $\Gamma(1013)/\Gamma(385) = 5.2$ also in good agreement with the corresponding intensity ratio measured in all our films.

Table I

wave number	2085	1985	890-840	630	(cm^{-1})	
mode	Si-H ₂ stretch	Si-H stretch	Si-H ₂ , Si-H ₃ bending	Si-H, Si-H ₂ wagging		
wave number	1010	930-920	828	510	380	300 (cm^{-1})
mode	SiF_4 stretch	Si-F_3 ; Si-F_2 stretch	Si-F stretch	Si TO (F-in-duced)	SiF_4 bending	Si-F, Si-F ₂ wagging

The Si-F stretching frequencies for a number of methyl and fluorine substituted silanes cover a range from 1013 cm^{-1} in SiF_4 to 858 cm^{-1} in FSiH_2CH_3 [5]. When plotted versus the electronegativity sum of the substituted groups, these frequencies extrapolate to a mode at 810 cm^{-1} expected for the Si-F stretch frequency in a-Si [6]. The closest band observed in our spectra is that at 828 cm^{-1} which we ascribe consequently to that mode. This assignment agrees with that made by others [1,7,8]. The frequency of the Si-F wagging mode is estimated as 291 cm^{-1} from the stretching frequency using the ratio of the bond-stretching to bond-bending force constants from SiF_4 [4]. The band closest to this value at 300 cm^{-1} in Fig.(1) is therefore identified with that mode. The stretching modes for Si-F₂ and Si-F₃ configurations are expected to lie between those of SiF_4 and Si-F. The asymmetric line at 930 cm^{-1} is thus the most likely candidate.

Shimada et al. [8] assign a prominent line at 1015 cm^{-1} to the asymmetric doubly degenerate Si-F₂ stretching band. They position the corresponding symmetric band at 838 cm^{-1} . We do not find lines

The hydrogen induced bands are readily identified in analogy with the corresponding absorption peaks in hydrogenated a-Si (a-Si:H) [2]. The frequencies are shifted downwards by 15 cm^{-1} (stretching) and 10 cm^{-1} (wagging) in the fluorinated samples. The remaining lines are fluorine related vibrations. The bands at 1010 cm^{-1} and at 380 cm^{-1} always occur together with a constant intensity ratio $I(1010)/I(380) = 5.0 \pm 0.5$. We ascribe them to the bond-stretching (1010 cm^{-1}) and bond-bending (380 cm^{-1}) vi-

at either of these positions. But if we suppose that our lines at 1010 cm^{-1} and 828 cm^{-1} are identical with their Si-F_3 bands we expect an intensity ratio of two for these lines on the basis of their degeneracy. Since a band at 827 cm^{-1} attributed to Si-F_2 by Shimada et al. could conceivably contribute to our line at 828 cm^{-1} as well, this ratio would be an upper limit. In Fig.(2) spectra are shown with an intensity ratio $I(1010)/I(828) > 2$ which exclude the interpretation of the 1010 cm^{-1} line as due to Si-F_3 .

Finally, the peak in the transmission spectra of Fig.(1)³ at 510 cm^{-1} lies within the TO band of the a-Si lattice (for c-Si the peak in the density of TO states occurs at 490 cm^{-1}). While this mode is ir-inactive in the non-polar Si network, the addition of the highly electronegative F should induce enough charge transfer in its neighborhood to ir-activate this mode in a-Si:F and shift its frequency to 510 cm^{-1} . The intensity of this band scales with the Si-F and Si-F_2 vibrational modes but not with the SiF_4 modes.

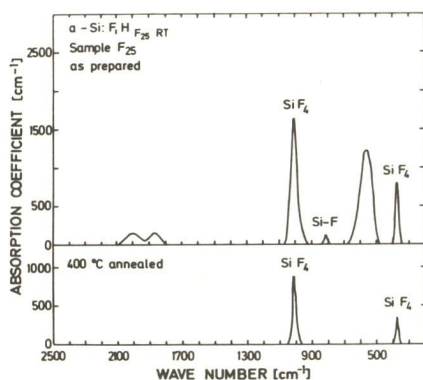


Fig. 2

Ir-spectra of $\text{a-Ge}_x\text{Si}_{1-x}\text{F}$ before and after annealing

absorption strengths of various ir bands as a function of T_a . The hydrogen bands show the annealing behaviour observed earlier in non-fluorinated a-Si:H samples [9].

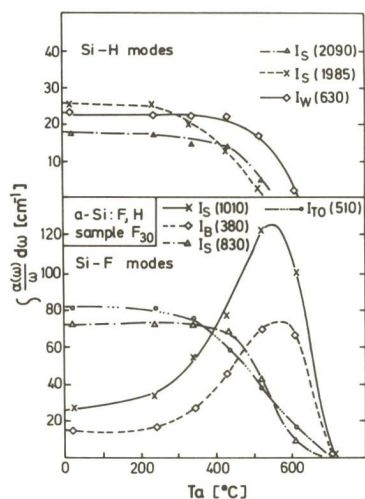


Fig. 3

Intensity of ir-absorption bands as a function of annealing temperature

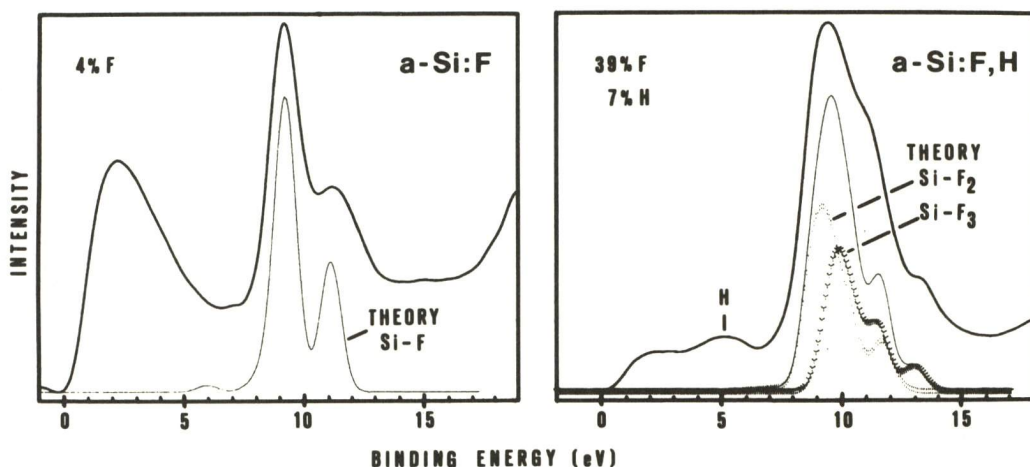


Fig. 4

Photoemission spectra of two fluorinated a-Si films

The photoemission spectra ($h\nu = 40.8$ eV) for two films containing 4%F and 39%F are shown in Fig.(4). The film with 4%F exhibits a doublet of F 2p induced states at 9.1 and 11.2 eV below E_F . The corresponding local density of states (LDOS) for a Si-F configuration calculated by Ching [10] is seen to reproduce the experimental result quite nicely after a rigid shift of 2.7 eV to lower binding energies has been applied. The 39%F film has a three peaked structure with 9.3, 11.2, and 13.2 eV binding energies. A superposition of the calculated Si-F₂ and Si-F₃ LDOS is again in good agreement with our results with the same 2.7 eV shift. From the position of the Fermi level in our spectra we conclude that the a-Si:F films are p-type. The top of the valence bands is apparently much less affected by the incorporation of F than by the addition of H [11].

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